

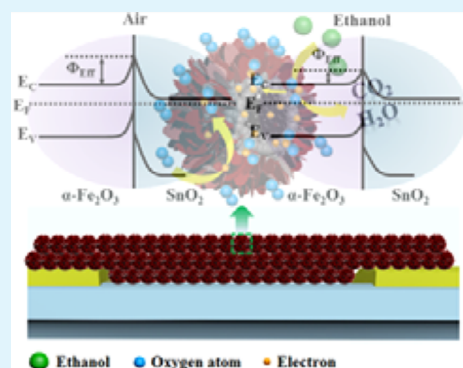
# Hierarchical Assembly of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> Nanosheets on SnO<sub>2</sub> Hollow Nanospheres with Enhanced Ethanol Sensing Properties

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## S Supporting Information

**ABSTRACT:** We present the preparation of a hierarchical nanoheterostructure consisting of inner SnO<sub>2</sub> hollow spheres (SHS) surrounded by an outer  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheet (FNS). Deposition of the FNS on the SHS outer surface was achieved by a facile microwave hydrothermal reaction to generate a double-shell SHS@FNS nanostructure. Such a composite with novel heterostructure acted as a sensing material for gas sensors. Significantly, the hierarchical composites exhibit excellent sensing performance toward ethanol, which is superior to the single component (SHS), mainly because of the synergistic effect and heterojunction.



**KEYWORDS:** microwave hydrothermal method, semiconductor, hollow nanostructure,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites, gas sensor

## 1. INTRODUCTION

Gas sensors have been acknowledged as profilers to detect and quantify inflammable, explosive, or toxic gases.<sup>1–4</sup> For the design and fabrication of high performance gas sensors in terms of sensitivity, selectivity, and stability, understanding the properties of the sensing material that affect sensor performance is of great necessary. Therefore, the development of highly efficient sensitive materials is one of the most important topics in gas sensor research. In general, ideal sensing materials for gas sensors should be cheap, nontoxic, and stable. Considering their advantages of high sensitivity and simplicity in synthesis, semiconductor oxides, including  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, SnO<sub>2</sub>, ZnO, and In<sub>2</sub>O<sub>3</sub>, have been increasingly researched as sensing materials over the past few decades.<sup>5–9</sup> The sensing mechanism of sensors employing semiconductor oxides is mainly that the adsorption and reaction of oxygen and target gases result in a change in the electrical conductivity.<sup>10–13</sup> Thus, the morphology, microstructure, composition, and crystalline size play an important role in their gas sensing properties. The sensitivity of the sensor increases significantly when the size of the sensing materials decrease to be a similar thickness as that of the electron depletion layer.<sup>14</sup> Many satisfactory results regarding sensing properties of semiconductor oxides have been obtained, but the design and development of highly sensitive and selective sensing materials remains a challenge for gas sensors. Indeed, during the past decade, many efforts have been dedicated to modifying semiconductor oxide sensing materials, including the modulation of surface states by doping with elements,<sup>15–17</sup> the construction of heterojunctions by combining them with different semiconductor oxides,<sup>18–20</sup> and the

addition of catalysts on the surface of the semiconductor oxides.<sup>21–23</sup> Recently, much research has been carried out aiming to enhance the performance of gas sensors by applying heterostructure semiconductor composites.<sup>24–26</sup> Because the sensing properties of heterostructure composites are strongly related to their morphologies, preparation of composites with unique architectures will be increasingly important for constructing high performance gas sensors.

SnO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, two of the important n-type semiconductors, have received widespread attention because of their unique characters and vast potential applications in various fields.<sup>27–31</sup> It has been previously demonstrated that an  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> hybrid composite can yield enhanced functions, such as gas response, photocatalyst, and lithium storage capacity.<sup>32–37</sup> Therefore, hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites with various architectures have been prepared through different strategies. Solution-phase synthesis methods have been demonstrated to be a versatile and efficient route to finely adjust semiconductor composites with different compositions and microstructures.<sup>38–40</sup> However, despite the exciting results that have been obtained, there are few reports on the hierarchical double-shell  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> nanostructures consisting of inner SnO<sub>2</sub> hollow nanospheres surrounded by outer  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheets. In this work, we successfully prepared hollow  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites by combining a hydrothermal route (for the hollow SnO<sub>2</sub> nanosphere) and a

Received: May 30, 2015

Accepted: August 18, 2015

Published: August 18, 2015

microwave-assisted hydrolytic reaction (for the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheets). Moreover, such novel  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> heterostructures as sensing materials have been demonstrated, and they displayed that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites had superior sensing performance to that of the SnO<sub>2</sub> individual component in terms of high gas response to ethanol. The variation of the heterojunction barrier at the diverse gas ambience and the synergistic effect of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub> were suggested to be the origin of the enhanced performance.

## 2. EXPERIMENTAL SECTION

**Preparation of Hollow  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> Hierarchical Composites.** SnO<sub>2</sub> hollow nanospheres were obtained as described in our previous publication.<sup>19</sup> The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> semiconductor heterostructures were synthesized through a microwave-assisted hydrothermal reaction without any surfactant. Briefly, 0.03 g of the as-prepared hollow SnO<sub>2</sub> spheres were added to a mixed solvent of 5 mL of glycerol and 35 mL of deionized water with vigorous stirring. Then, 0.79 g of FeCl<sub>2</sub>·4H<sub>2</sub>O and 0.69 g of K<sub>2</sub>SO<sub>4</sub> were dissolved into the mixed solution, respectively. Then, the above suspension was poured into a Teflon-lined autoclave and heated with a predetermined heating procedure (first heated to 180 °C from 25 °C, then kept at 180 °C for 2 h, and afterwards cooled naturally) in a microwave hydrothermal apparatus. The resulting product was centrifuged and washed with deionized water and ethanol several times and finally dried at 80 °C.

**Characterization.** The phase structures of the as-obtained samples were characterized through X-ray powder diffraction on a Rigaku TTRIII X-ray diffractometer, and the data were recorded from 20° to 70°. The morphologies and microstructures were investigated by field-emission scanning electron microscopy (FESEM) using a JEOL JSM-7500F microscope. Transmission electron microscopy (TEM) images and the energy dispersive X-ray spectrometry (EDS) pattern were obtained by a JEOL JEM-3010 microscope and its attachment, respectively. Specific surface areas of the samples were calculated by the Brunauer–Emmett–Teller (BET) equation according to the N<sub>2</sub> adsorption/desorption isotherm recorded using Micromeritics Gemini VII equipment (Surface Area and Porosity System).

**Fabrication and Measurement of Gas Sensors.** For gas response, the obtained products were coated onto the surface of a commercially produced ceramic tube to form a thick sensing film. The as-fabricated devices were fired to 400 °C for 2 h in air using a muffle furnace. A Ni–Cr alloy coil, as a heater, was threaded to the ceramic tube to regulate the entire operating temperature of the sensor. The details of the fabrication and measurement process of the gas sensor are described in our previous works.<sup>15,23</sup> A static system was applied to investigate the gas sensing performances of the sensors under laboratory conditions (40% RH, 23 °C). The response of the sensor was measured as the resistance ratio between the air ( $R_a$ ) and the target gas ( $R_g$ ). The time consumed by the sensor to reach 90% of the total resistance change in tested gas and air were defined as the response time and recovery time, respectively.

## 3. RESULTS AND DISCUSSION

**Structural and Morphological Characteristics.** The XRD pattern in Figure 1 displays the phase purity and crystallographic structure of the as-prepared hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> nanostructures. All the diffraction peaks in this XRD pattern for the composite contained a tetragonal rutile structure of SnO<sub>2</sub> and rhombohedral structure of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, which were accorded well with those from the standard Joint Committee on Powder Diffraction Standards (JCPDS) card of SnO<sub>2</sub>, No. 41-1445, with  $a = 0.4738$  nm and  $c = 0.3187$  nm and the JCPDS card of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, No. 33-0664, with  $a = 0.5035$  nm and  $c = 1.375$  nm, respectively. Moreover, no other diffraction peaks derived from impurities were found, indicating high purity of the as-prepared product.

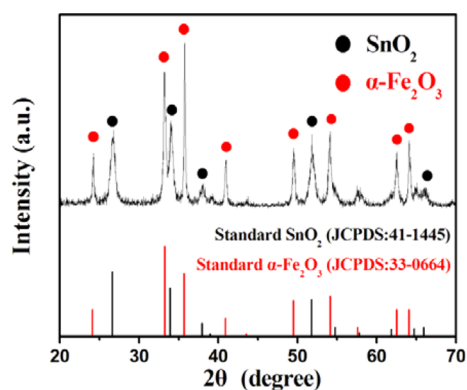


Figure 1. XRD pattern of the hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites.

The images in Figure 2 present the morphologies and microstructures of the as-obtained samples: hollow SnO<sub>2</sub>

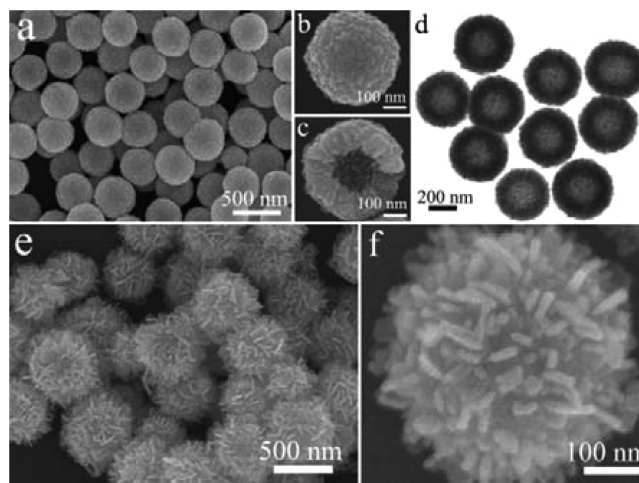
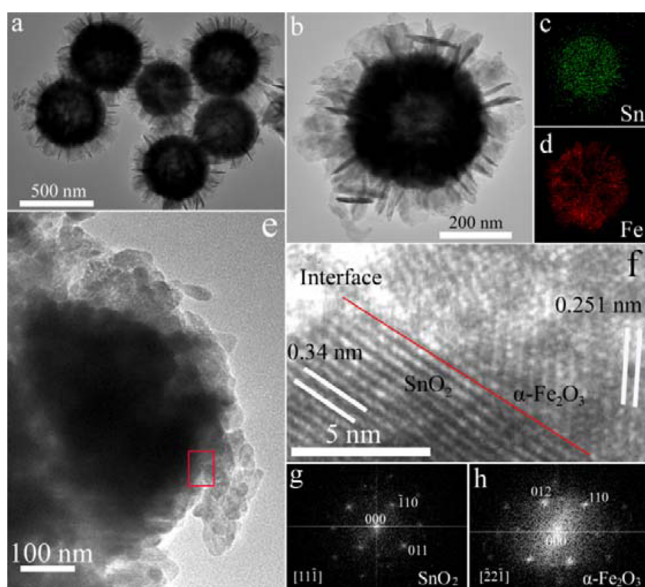


Figure 2. (a–d) Typical FESEM and TEM images of SnO<sub>2</sub> spheres. (e and f) FESEM images of the as-synthesized hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites.

nanospheres and hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> nanostructures. Figure 2a depicts a low-magnification FESEM image of the SnO<sub>2</sub> sample fabricated through the first hydrothermal route using SnC<sub>2</sub>O<sub>4</sub> as the precursor. It is apparent that the product consisted of monodisperse spheres that were ~400 nm in diameter. No other morphologies could be observed, which indicated that good uniformity and dispersibility were achieved through this strategy. The enlarged FESEM image in Figure 2b provides a single SnO<sub>2</sub> sphere; it is found that the SnO<sub>2</sub> sphere consisted of many primary particles with a size of a few dozen nanometers. From a broken nanosphere, as shown in Figure 2c, the hollow structure and the nanometer-scale primary particles could be clearly identified. Moreover, the thickness of spherical shell was ~100 nm. TEM images further reveal the hollow interior of the SnO<sub>2</sub> spheres. As can be seen in Figure 2d, the contrasting light and dark between the center and the edge of the spheres confirmed that these spheres possessed a hollow interior. Figure S1 displays a high-resolution TEM (HRTEM) image of nanoparticles on the surface of the sphere where the (110) lattice spacing of ~0.335 nm of tetragonal rutile SnO<sub>2</sub> could be clearly observed.<sup>41</sup> After the second solution growth, the product exhibited hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> hollow nanostructures. A panoramic view of the composites, as

shown in Figure 2e, reveals that they were composed of very uniform spherical particles with diameters of  $\sim 600$  nm. The high-magnification FESEM presented in Figure 2f displays the detailed morphological information on a single hierarchical sphere. It is found that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheets, with an average thickness of  $\sim 10$  nm, assembled hierarchically on the surface of hollow SnO<sub>2</sub> nanospheres.

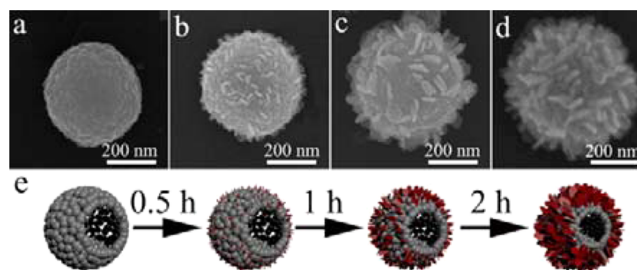
Such hierarchical composites were analyzed by TEM to get deeper insight into their structure, images of which are displayed in Figure 3. An obviously hollow interior could be



**Figure 3.** (a, b, and e) Typical TEM images of as-obtained hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composites. (c and d) Elemental mapping images. (f) HRTEM images of the interfacial region if  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> hollow nanosphere. (g and h) FFT patterns taken from SnO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, respectively.

clearly observed from a low magnification TEM image (Figure 3a). Careful observation of a single sphere (Figure 3b) indicates that the hollow sphere had an obvious double-shell structure. The inner shell could be identified as a hollow SnO<sub>2</sub> sphere, which appeared black colored in the image due to high mass–thickness contrast. The outer shell of the hierarchical sphere was composed of numerous  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheets. The thickness of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> shell was observed to be  $\sim 100$  nm by measuring the light-colored region in the image. The TEM elemental mapping taken from a single hierarchical sphere (Figure 3b) clearly identified the spatial distributions of Sn and Fe elements in the composite, as shown in Figure 3c and d. The signal of Sn was strongly detected in the inner shell region of the hollow sphere, whereas the Fe element was mainly distributed in the outer shell region. A typical HRTEM image of the interfacial region in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> hollow nanospheres (outlined by a red box in Figure 3e) is presented in Figure 3f, which indicated that there was good lattice compatibility at the interface. The fringe spacings were approximately 0.34 and 0.251 nm, which corresponded to the (110) plane of SnO<sub>2</sub> and the (110) plane of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, respectively.<sup>30,41</sup> The fast-Fourier transform (FFT) patterns selected from SnO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are displayed in Figure 3g and h, respectively. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanosheet gave the (012) plane to constitute the interface with the (110) plane of SnO<sub>2</sub>.<sup>33</sup>

The formation mechanism of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> heterostructure composite was investigated through a series of experiments. The FESEM images in Figure 4a–d show the

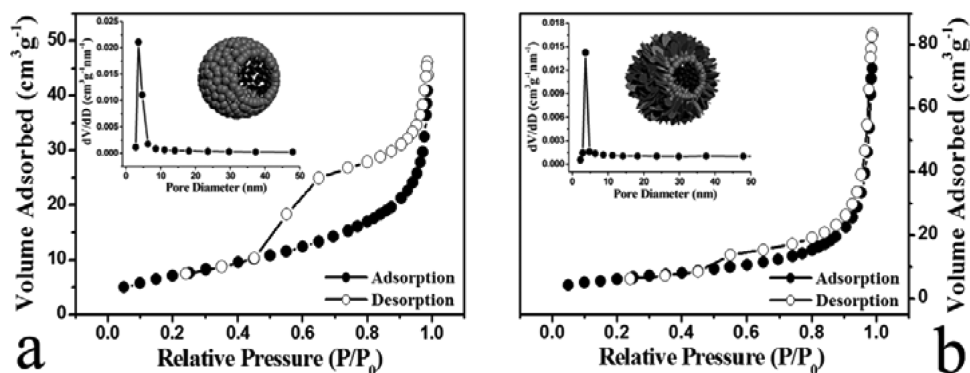


**Figure 4.** (a–d) FESEM images of the samples at various reaction times: (a) 0 h, (b) 0.5 h, (c) 1 h, and (d) 2 h. (e) Schematic of the growth process of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> composite.

samples prepared at different reaction times, which reveal the morphological and structural changes from SnO<sub>2</sub> hollow spheres to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> double-shell structural composites. During the initial 30 min reaction, small nanosheets of  $\sim 8$  nm thickness were found to be growing on the surfaces of the SnO<sub>2</sub> hollow spheres (Figure 4b). With an extended reaction time, the size, thickness, and quantity of nanosheets increased, and the product exhibited a flower-like morphology (Figure 4c and d). On the basis of the above observations, a process of forming the hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> double-shell hollow spheres during the microwave hydrothermal reaction is proposed and is presented schematically in Figure 4e. At first, the hydrolysis of FeCl<sub>2</sub>·4H<sub>2</sub>O will lead to the formation of iron oxide nanoparticles; then, these nanoparticles will aggregate on the surface of SnO<sub>2</sub> spheres. During the extended microwave hydrothermal reaction, the as-obtained nanoparticles will grow into flaky nanostructures, forming rough shells, which result in the obtention of a double-shell hollow structure. The specific growth mechanism of the unique double-shell heterostructures is still under investigation. Presented here is just an assumption that corresponds well with the results from electron microscopy.

The specific surface areas of hollow SnO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> spheres were estimated through the data of N<sub>2</sub> adsorption/desorption isotherms by the BET method, as shown in Figure 5. For two structures, an obvious hysteresis loop was observed from the nitrogen adsorption isotherm, which indicated mesoporous structures in hollow spheres.<sup>42</sup> From the corresponding pore size distribution curves (insets in Figure 5), the distribution centers had almost no change. Moreover, as found by the results of the BET measurements, such a hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> heterostructure composite gave a slightly larger specific surface area (26.1 m<sup>2</sup> g<sup>-1</sup>) and total pore volume (0.13 cm<sup>3</sup> g<sup>-1</sup>) relative to those of the hollow SnO<sub>2</sub> spheres (22.9 m<sup>2</sup> g<sup>-1</sup>, 0.07 cm<sup>3</sup> g<sup>-1</sup>, respectively). The relatively high pore volume in the double shell  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> hollow spheres seems to afford an efficient way to transmit to their interior space, which is critical for gas sensing and other applications.

**Sensing Properties.** Recently, vast efforts have been spent searching for excellent sensing materials to enhance the performance of gas sensors. Here, we employed the prepared hierarchical  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> nanostructures as a potential sensing material for gas sensors as compared with the pristine SnO<sub>2</sub> hollow nanospheres. For semiconductor oxide gas



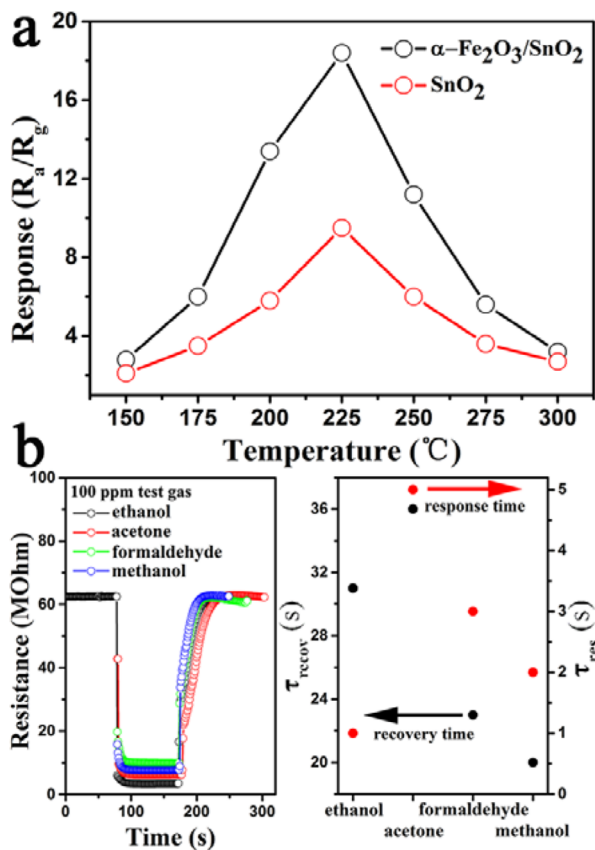
**Figure 5.** Typical  $N_2$  adsorption/desorption isotherms and pore size distribution curves of (a)  $SnO_2$  hollow nanospheres and (b) hierarchical  $\alpha-Fe_2O_3/SnO_2$  composites.

sensors, the operating temperature plays an important role in their response to target gas.<sup>10</sup> Therefore, the responses of the two sensors to 100 ppm of  $C_2H_5OH$  were measured as a function of operating temperature to ascertain the optimum operating temperature of the sensors, as shown in Figure 6a. From which a volcano-like correlation between response and operating temperature is observed. In other words, the responses of the tested sensors increased first and then decreased with increasing temperature. The response showed a maximum value of 18.4 for the hierarchical  $\alpha-Fe_2O_3/SnO_2$  composites and 9.5 for the hollow  $SnO_2$  nanospheres at 225

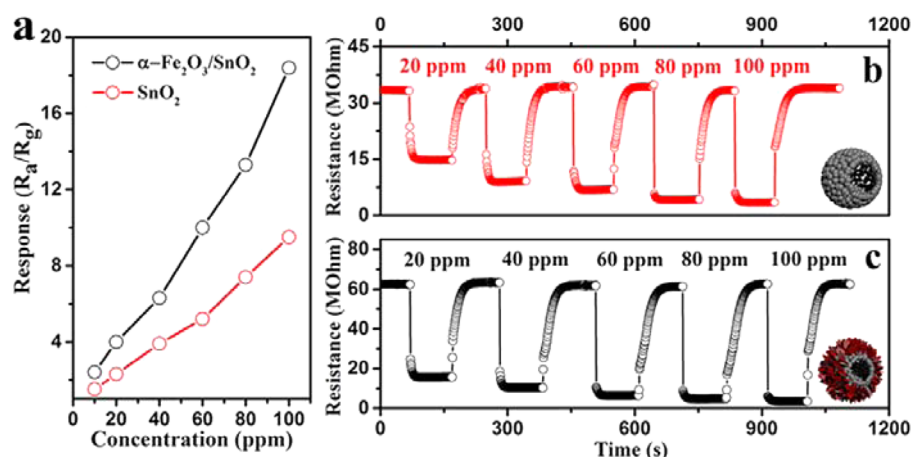
$^{\circ}C$ , which was therefore chosen for the two gas sensors as the optimal operating temperature. The results indicate that the hierarchical composites displayed a remarkably improved gas response (18.4), which was almost twice that of the  $SnO_2$  hollow spheres (9.5).

The dynamic response characteristics of the  $\alpha-Fe_2O_3/SnO_2$  sensor to diverse reducing gases were then studied. Figure 6b presents the response of hollow heterostructures to 100 ppm of  $C_2H_5OH$ ,  $C_3H_6O$ , HCHO, and  $CH_3OH$  for a cycle period at 225  $^{\circ}C$ . It is found that the sensor showed an n-type response to tested gases and exhibited a sensitive and reversible response. Obviously, with the introduction of tested gases, the sensor responded immediately and the resistance reached a near steady state within seconds.<sup>43</sup> Because the tested gas further diffused through the sensing layer and occupied the residual surface reaction sites, the resistance then changed slowly. When the sensor was placed in air, the resistance gradually increased and finally returned to almost its baseline level. The response was 18.4, 10.1, 6.3, and 8.1 for 100 ppm of  $C_2H_5OH$ ,  $C_3H_6O$ , HCHO, and  $CH_3OH$ , respectively. The reproducibility of the sensor upon 8 cycle response measurements to 60 ppm ethanol at 225  $^{\circ}C$  (Figure S2) revealed there was no evident change in the response amplitude, which indicated that the sensor had good reproducibility. Moreover, among these tested gases, the sensor showed the maximum response to ethanol. The response time of the sensor was 1, 5, 3, and 2 s for ethanol, acetone, formaldehyde, and methanol, respectively. After exposure in air, the recovery time of the sensor was 31, 36, 23, and 20 s for ethanol, acetone, formaldehyde, and methanol, respectively.

The relationships between the responses and ethanol concentrations for the two sensors at 225  $^{\circ}C$  is depicted in Figure 7a. From the curves, it can be seen that the responses of the sensors were proportional to the increasing concentrations of ethanol from 10 to 100 ppm. However, the sensor, using hierarchical composites, by contrast, exhibited a high response to each concentration of ethanol. Moreover, the response of the  $\alpha-Fe_2O_3/SnO_2$  composites tended to rise faster than that of the hollow  $SnO_2$  spheres with increasing  $C_2H_5OH$  concentration. Panels b and c in Figure 7 demonstrate the dynamic resistance changes of the two sensors to various concentrations of ethanol at 225  $^{\circ}C$ . Obviously, the response and recovery characteristics of the sensors were nearly reproducible with the rapid response and recovery. The response of the sensor using heterostructure  $\alpha-Fe_2O_3/SnO_2$  hierarchical composites was approximately 2.4, 4.1, 6.3, 9.9, 13.3, and 18.4 to 10, 20, 40, 60, 80, and 100 ppm of  $C_2H_5OH$ , respectively, whereas the response of pure  $SnO_2$



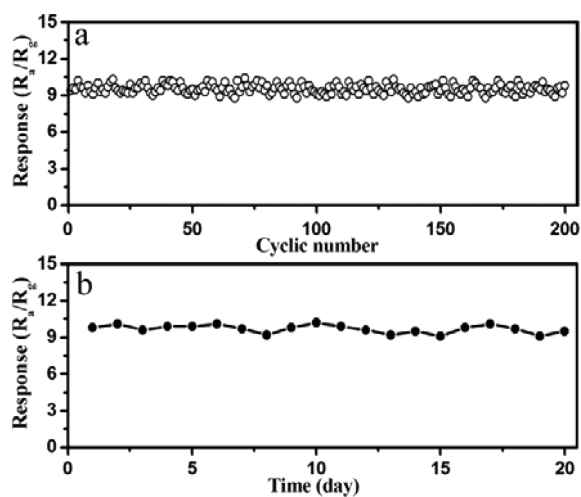
**Figure 6.** (a) Response of pristine  $SnO_2$  hollow nanospheres and  $\alpha-Fe_2O_3/SnO_2$  hierarchical composites relative to operating temperature to 100 ppm of  $C_2H_5OH$ . (b) Response transients of hollow  $\alpha-Fe_2O_3/SnO_2$  double-shell heterostructures to 100 ppm of  $C_2H_5OH$ ,  $C_3H_6O$ , HCHO, and  $CH_3OH$  at 225  $^{\circ}C$ .



**Figure 7.** (a) Responses of the two sensors relative to  $C_2H_5OH$  concentrations at 225 °C. (b and c) Response transients of hollow  $SnO_2$  nanospheres and hierarchical  $\alpha-Fe_2O_3/SnO_2$  heterostructures to diverse concentrations of ethanol at 225 °C.

was 1.5, 2.3, 3.9, 5.2, 7.4, and 9.5 for each respective concentration.

In terms of practical applications, the long-term stability of the sensor is considered an important parameter. Therefore, successive tests of the response toward 60 ppm ethanol for the sensor using the as-obtained hierarchical  $\alpha-Fe_2O_3/SnO_2$  heterostructures at 225 °C were carried out, as shown in Figure 8a. It is found that the sensor maintained its initial



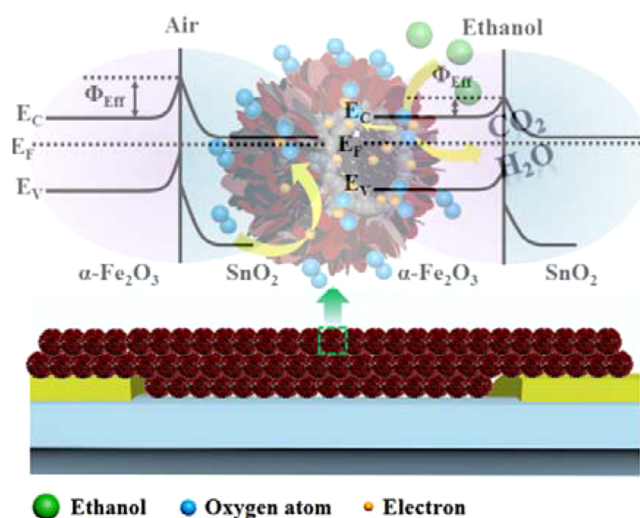
**Figure 8.** Response of hierarchical  $\alpha-Fe_2O_3/SnO_2$  hollow heterostructures as a function of the number of (a) cyclic measurements and (b) tested days to 60 ppm of ethanol at 225 °C.

response after 200 cyclic tests, which indicated that the hierarchical composites had excellent repeatability. Moreover, the repeated tests were conducted every day, which revealed that the responses were almost constant without an obvious increase/decrease even after 20 days, as shown in Figure 8b. The result further confirmed that the present sensor might have a practical application on the basis of its good stability.

The enhanced performance is more likely derived from the novel heterostructure of the as-synthesized composite rather than simply a result of the addition of the other sensing component, which is described as follows. On one hand, the improved response of hierarchical composites compared with the primary  $SnO_2$  hollow nanospheres can be easily understood by the synergistic effect of  $\alpha-Fe_2O_3$  and  $SnO_2$ . It has been

reported that the surface reactions between tested gases and adsorbed oxygen species depend on the acid–base properties of oxide semiconductors. In terms of ethanol gas, the base property will be beneficial for the reaction with adsorbed oxygen.<sup>44–48</sup> In this work, the addition of  $\alpha-Fe_2O_3$  enhances the basicity of the composite because  $\alpha-Fe_2O_3$  and  $SnO_2$  are basic oxides. Therefore, the heterostructure of the composites exhibits a higher response to ethanol than that of pristine  $SnO_2$ . On the other hand, the band configuration at the interface of the  $\alpha-Fe_2O_3/SnO_2$  composites in different atmospheres is proposed, as shown in Scheme 1, where  $\Phi_{eff}$

**Scheme 1.** Schematic Diagram of Band Configuration at the Interface of the  $\alpha-Fe_2O_3/SnO_2$  Nanoheterostructure in Different Atmospheres



is the effective barrier height.<sup>49</sup> For composites, electron transport will be strongly modulated by the heterojunction barrier, which has been generally researched for many heterostructures of devices.<sup>50–53</sup> It has been clarified that the conductivity of the heterostructures of the composites is inversely proportional to the barrier height at the high temperature.<sup>45</sup> When the composites are exposed to air, the electrons in conduction bands of  $\alpha-Fe_2O_3$  and  $SnO_2$  are trapped by oxygen to form adsorbed oxygen species, which results in the increase of the barrier height (Scheme 1), thereby

decreasing the conductivity owing to the lower free electron concentration upon exposure to a given reducing gas, such as  $C_2H_5OH$ , which will react with oxygen species. As a result, the electrons trapped in the adsorbed oxygen species are released back into the conduction bands of  $\alpha-Fe_2O_3$  and  $SnO_2$ .<sup>54,55</sup> As such, the barrier height will be decreased due to the higher concentration of electrons. Therefore, the resistance of the heterostructures is greatly decreased. This means that the response of hierarchical composites to ethanol can be enhanced. Thus, the synergistic effect and the change of barrier height in the diverse gas atmosphere may be the origin of the improvement in gas sensing performance.

#### 4. CONCLUSIONS

In summary, a novel hierarchical  $\alpha-Fe_2O_3/SnO_2$  nanoheterostructure has been prepared by a facile solution strategy. The uniqueness of the structure is that  $\alpha-Fe_2O_3$  nanosheets are hierarchically assembled onto the surface of  $SnO_2$  hollow nanospheres. The as-obtained  $\alpha-Fe_2O_3/SnO_2$  composite was applied as the sensing material for the gas sensor. In comparison to a single component ( $SnO_2$  hollow nanosphere), the composite has demonstrated superior sensing performance toward ethanol. The improvement may be ascribed to variation of the heterojunction barrier in the diverse gas atmosphere.

#### ■ ASSOCIATED CONTENT

##### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsami.5b04751.

Typical TEM images of as-prepared hollow  $SnO_2$  spheres, HRTEM images, and response transitions of the sensor using double-shell  $\alpha-Fe_2O_3/SnO_2$  hollow spheres to 60 ppm of ethanol (PDF)

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##### Notes

The authors declare no competing financial interest.

#### ■ ACKNOWLEDGMENTS

This work is supported by the National Nature Science Foundation of China (Nos. 61374218, 61134010, and 61327804), Program for Chang Jiang Scholars and Innovative Research Team in University (No. IRT13018), and National High-Tech Research and Development Program of China (863 Program, No. 2013AA030902 and 2014AA06A505).

#### ■ ABBREVIATIONS

SHS,  $SnO_2$  hollow spheres  
FNS,  $\alpha-Fe_2O_3$  nanosheets  
XRD, X-ray power diffraction  
FESEM, field-emission scanning electron microscopy  
TEM, transmission electron microscopy  
HRTEM, high-resolution transmission electron microscopy  
EDS, energy dispersive X-ray spectrometry  
FFT, fast-Fourier transform  
BET, Brunauer–Emmett–Teller

#### ■ REFERENCES

- (1) Kong, J.; Franklin, N. R.; Zhou, C.; Chapline, M. G.; Peng, S.; Cho, K.; Dai, H. Nanotube Molecular Wires as Chemical Sensors. *Science* **2000**, *287*, 622–625.
- (2) Hahn, Y. B.; Ahmad, R.; Tripathy, N. Chemical and Biological Sensors Based on Metal Oxide Nanostructures. *Chem. Commun.* **2012**, *48*, 10369–10385.
- (3) Cui, Y.; Wie, Q. Q.; Park, H. K.; Lieber, C. M. Nanowire Nanosensors for Highly Sensitive and Selective Detection of Biological and Chemical Species. *Science* **2001**, *293*, 1289–1292.
- (4) Hagleitner, C.; Hierlemann, A.; Lauge, D.; Kimmer, A.; Kerness, N.; Brand, O.; Baltes, H. Smart Single-Chip Gas Sensor Microsystem. *Nature* **2001**, *414*, 293–296.
- (5) Wang, Y. L.; Jiang, X. C.; Xia, Y. N. A Solution-Phase, Precursor Route to Polycrystalline  $SnO_2$  Nanowires That Can Be Used for Gas Sensing Under Ambient Conditions. *J. Am. Chem. Soc.* **2003**, *125*, 16176–16177.
- (6) Hernandez-Ramirez, F.; Prades, J. D.; Hackner, A.; Fischer, T.; Mueller, G.; Mathur, S.; Morante, R. Miniaturized Ionization Gas Sensors from Single Metal Oxide Nanowires. *Nanoscale* **2011**, *3*, 630–634.
- (7) Vallejos, S.; Stoycheva, T.; Umek, P.; Navio, C.; Snyders, R.; Bittencourt, C.; Llobet, E.; Blackman, C.; Moniz, S.; Correig, X. Au Nanoparticle-Functionalised  $WO_3$  Nanoneedles and Their Application in High Sensitivity Gas Sensor Devices. *Chem. Commun.* **2011**, *47*, 565–567.
- (8) Waitz, T.; Wagner, T.; Sauerwald, T.; Kohl, C.-D.; Tiemann, M. Ordered Mesoporous  $In_2O_3$ : Synthesis by Structure Replication and Application as a Methane Gas Sensor. *Adv. Funct. Mater.* **2009**, *19*, 653–661.
- (9) Li, J.; Fan, H. Q.; Jia, X. H. Multilayered ZnO Nanosheets with 3D Porous Architectures: Synthesis and Gas Sensing Application. *J. Phys. Chem. C* **2010**, *114*, 14684–14691.
- (10) Yamazoe, N.; Sakai, G.; Shimanoe, K. Oxide Semiconductor Gas Sensor. *Catal. Surv. Asia* **2003**, *7*, 63–75.
- (11) Egashira, M.; Shimizu, Y.; Takao, Y.; Sako, S. Variations in I-V Characteristics of Oxide Semiconductors Induced by Oxidizing Gases. *Sens. Actuators, B* **1996**, *35*, 62–67.
- (12) Dayan, N. J.; Sainkar, S. R.; Karekar, R. N.; Aiyer, R. C. Formulation and Characterization of ZnO:Sb Thick-Film Gas Sensor. *Thin Solid Films* **1998**, *325*, 254–258.
- (13) Williams, D. E. Semiconductor Oxides as Gas-Sensitive Resistors. *Sens. Actuators, B* **1999**, *57*, 1–16.
- (14) Yamazoe, N. New Approaches for Improving Semiconductor Gas Sensors. *Sens. Actuators, B* **1991**, *5*, 7–19.
- (15) Sun, P.; Zhou, X.; Wang, C.; Wang, B.; Xu, X. M.; Lu, G. Y. One-Step Synthesis and Gas Sensing Properties of Hierarchical Cd-Doped  $SnO_2$  Nanostructures. *Sens. Actuators, B* **2014**, *190*, 32–39.
- (16) Wang, H. K.; Dou, K. P.; Teoh, W. Y.; Zhan, Y. W.; Hung, T. F.; Zhang, F. H.; Xu, J. Q.; Zhang, R. Q.; Rogach, A. L. Engineering of Facets, Band Structure, and Gas Sensing Properties of Hierarchical  $Sn^{2+}$ -Doped  $SnO_2$  Nanostructures. *Adv. Funct. Mater.* **2013**, *23*, 4847–4853.
- (17) Han, N.; Wu, X. F.; Zhang, D. W.; Shen, G. L.; Liu, H. D.; Chen, Y. F. CdO Activated Sn-Doped ZnO for Highly Sensitive, Selective and Stable Formaldehyde Sensor. *Sens. Actuators, B* **2011**, *152*, 324–329.
- (18) Shan, H.; Liu, C. B.; Liu, L.; Zhang, J. B.; Li, H. Y.; Liu, Z.; Zhang, X. B.; Bo, X. Q.; Chi, X. Excellent Toluene Sensing Properties of  $SnO_2-Fe_2O_3$  Interconnected Nanotubes. *ACS Appl. Mater. Interfaces* **2013**, *5*, 6376–6380.
- (19) Sun, P.; Zhou, X.; Wang, C.; Shimanoe, K.; Lu, G. Y.; Yamazoe, N. Hollow  $SnO_2/\alpha-Fe_2O_3$  Spheres with a Double-Shell Structure for Gas Sensors. *J. Mater. Chem. A* **2014**, *2*, 1302–1308.
- (20) Si, S. F.; Li, C. H.; Wang, X.; Peng, Q.; Li, Y. D.  $Fe_2O_3/ZnO$  Core-Shell Nanorods for Gas Sensors. *Sens. Actuators, B* **2006**, *119*, 52–56.
- (21) Lü, R. J.; Zhou, W.; Shi, K. Y.; Yang, Y.; Wang, L.; Pan, K.; Tian, C. G.; Ren, Z. Y.; Fu, H. G. Alumina Decorated  $TiO_2$  Nanotubes with

Ordered Mesoporous Walls as High Sensitivity NO<sub>x</sub> Gas Sensor at Room Temperature. *Nanoscale* **2013**, *5*, 8569–8576.

(22) Kolmakov, A.; Kienov, D. O.; Liach, Y.; Stemmer, S.; Moskovits, M. Enhanced Gas Sensing by Individual SnO<sub>2</sub> Nanowires and Nanobelts Functionalized with Pd Catalyst Particles. *Nano Lett.* **2005**, *5*, 667–673.

(23) Sun, P.; Yu, Y. S.; Xu, J.; Sun, Y. F.; Ma, J.; Lu, G. Y. One-Step Synthesis and Gas Sensing Characteristics of Hierarchical SnO<sub>2</sub> Nanorods Modified by Pd Loading. *Sens. Actuators, B* **2011**, *160*, 244–250.

(24) Chu, D. W.; Zeng, Y. P.; Jiang, D. L.; Masuda, Y. In<sub>2</sub>O<sub>3</sub>-SnO<sub>2</sub> Nano-Toasts and Nanorods: Precipitation Preparation, Formation Mechanism, and Gas Sensitive Properties. *Sens. Actuators, B* **2009**, *137*, 630–636.

(25) Yu, H. L.; Li, L.; Gao, X. M.; Zhang, Y.; Meng, F. N.; Wang, T. S.; Xiao, G.; Chen, Y. J.; Zhu, C. L. Synthesis and H<sub>2</sub>S Gas Sensing Properties of Cage-Like MoO<sub>3</sub>/ZnO Composite. *Sens. Actuators, B* **2012**, *171/172*, 679–685.

(26) Zhang, S. F.; Ren, F.; Wu, W.; Zhou, J.; Xiao, X. H.; Sun, L. J.; Liu, Y.; Jiang, C. Z. Controllable Synthesis of Recyclable Core-Shell γ-Fe<sub>2</sub>O<sub>3</sub>@SnO<sub>2</sub> Hollow Nanoparticles with Enhanced Photocatalytic and Gas Sensing Properties. *Phys. Chem. Chem. Phys.* **2013**, *15*, 8228–8236.

(27) Huang, H.; Ong, C. Y.; Guo, J.; White, T.; Tse, M. S.; Tan, O. K. Pt Surface Modification of SnO<sub>2</sub> Nanorod Arrays for CO and H<sub>2</sub> Sensors. *Nanoscale* **2010**, *2*, 1203–1207.

(28) Yao, F.; Duong, D. L.; Lim, S. L.; Yang, S. B.; Hwang, H. R.; Yu, W. J.; Günes, F.; Lee, Y. H. Humidity-Assisted Selective Reactivity Between NO<sub>2</sub> and SO<sub>2</sub> Gas on Carbon Nanotubes. *J. Mater. Chem.* **2011**, *21*, 4502–4508.

(29) Ding, D. J.; Lou, X. W. SnO<sub>2</sub> Nanosheet Hollow Spheres with Improved Lithium Storage Capabilities. *Nanoscale* **2011**, *3*, 3586–3588.

(30) Cao, M. H.; Liu, T. F.; Gao, S.; Sun, G. B.; Wu, X. L.; Hu, C. W.; Wang, Z. L. Single Crystal Dendritic Micro-Pines of Magnetic α-Fe<sub>2</sub>O<sub>3</sub>: Large-Scale Synthesis, Formation Mechanism, and Properties. *Angew. Chem., Int. Ed.* **2005**, *44*, 4197–4201.

(31) Li, K. M.; Li, Y. J.; Lu, M. Y.; Kuo, C.; Chen, L. J. Direct Conversion of Single-Layer SnO Nanoplates to Multi-Layer SnO<sub>2</sub> Nanoplates with Enhanced Ethanol Sensing Properties. *Adv. Funct. Mater.* **2009**, *19*, 2453–2456.

(32) Yu, Q. X.; Zhu, J. H.; Xu, Z. Y.; Huang, X. T. Facile Synthesis of α-Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> Core-Shell Heterostructure Nanotubes for High Performance Gas Sensors. *Sens. Actuators, B* **2015**, *213*, 27–34.

(33) Niu, M. T.; Huang, F.; Cui, L. F.; Huang, P.; Yu, Y. L.; Wang, Y. S. Hydrothermal Synthesis, Structural Characteristics, and Enhanced Photocatalysis of SnO<sub>2</sub>/α-Fe<sub>2</sub>O<sub>3</sub> Semiconductor Nanoheterostructures. *ACS Nano* **2010**, *4*, 681–688.

(34) Wu, W.; Zhang, S. F.; Ren, F.; Xiao, X. H.; Zhou, J.; Jiang, C. Z. Controlled Synthesis of Magnetic Ion Oxides@SnO<sub>2</sub> Quasi-Hollow Core-Shell Heterostructures: Formation Mechanism, and Enhanced Photocatalytic Activity. *Nanoscale* **2011**, *3*, 4676–4684.

(35) Kang, J.; Kuang, Q.; Xie, Z. X.; Zheng, L. S. Fabrication of The SnO<sub>2</sub>/α-Fe<sub>2</sub>O<sub>3</sub> Hierarchical Heterostructure and Its Enhanced Photocatalytic Property. *J. Phys. Chem. C* **2011**, *115*, 7874–7879.

(36) Zhang, D. F.; Sun, L. D.; Jia, C. J.; Yan, Z. G.; You, L. P.; Yan, C. H. Hierarchical Assembly of SnO<sub>2</sub> Nanorod Arrays on α-Fe<sub>2</sub>O<sub>3</sub> Nanotubes: A Case of Interfacial Lattice Compatibility. *J. Am. Chem. Soc.* **2005**, *127*, 13492–13493.

(37) Romyantseva, M. N.; Kovalenko, V. V.; Gaskov, A. M.; Pagnier, T.; Machon, D.; Arbiol, J.; Morante, J. R. Nanocomposites SnO<sub>2</sub>/Fe<sub>2</sub>O<sub>3</sub>: Wet Chemical Synthesis and Nanostructure Characterization. *Sens. Actuators, B* **2005**, *109*, 64–74.

(38) Wang, B.; Chen, J. S.; Lou, X. W. The Comparative Lithium Storage Properties of Urchin-Like Hematite Spheres: Hollow vs. Solid. *J. Mater. Chem.* **2012**, *22*, 9466–9468.

(39) Zhou, L.; Zhao, D. Y.; Lou, X. W. Double-Shelled CoMn<sub>2</sub>O<sub>4</sub> Hollow Microcubes as High-Capacity Anodes for Lithium Ion Batteries. *Adv. Mater.* **2012**, *24*, 745–748.

(40) Kim, H.-R.; Choi, K.-I.; Kim, K.-M.; Kim, I.-D.; Cao, G. Z.; Lee, J.-H. Ultra-Fast Responding and Recovering C<sub>2</sub>H<sub>5</sub>OH Sensors Using SnO<sub>2</sub> Hollow Spheres Prepared and Activated by Ni Templates. *Chem. Commun.* **2010**, *46*, 5061–5063.

(41) Zhang, D. F.; Sun, L. D.; Yin, J. L.; Yan, C. H. Low-Temperature Fabrication of Highly Crystalline SnO<sub>2</sub> Nanorods. *Adv. Mater.* **2003**, *15*, 1022–1025.

(42) Gu, X.; Chen, L.; Ju, Z. C.; Xu, H. Y.; Yang, J.; Qian, Y. T. Controlled Growth of Porous α-Fe<sub>2</sub>O<sub>3</sub> Branches on β-MnO<sub>2</sub> Nanorods for Excellent Performance in Lithium-ion Batteries. *Adv. Funct. Mater.* **2013**, *23*, 4049–4056.

(43) Naisbitt, S. C.; Pratt, K. F. E.; Williams, D. E.; Parkin, I. P. A Microstructural Model of Semiconducting Gas Sensor Response: The Effects of Sintering Temperature on the Response of Chromium (CTO) to Carbon Monoxide. *Sens. Actuators, B* **2006**, *114*, 969–977.

(44) Jinkawa, T.; Sakai, G.; Tamaki, J.; Miura, N.; Yamazoe, N. Relationship Between Ethanol Gas Sensitivity and Surface Catalytic Property of Tin Oxide Sensors Modified with Acidic or Basic Oxides. *J. Mol. Catal. A: Chem.* **2000**, *155*, 193–200.

(45) Kotsikau, D.; Ivanovskaya, M.; Orlik, D.; Falasioni, M. Gas-Sensitive Properties of Thin and Thick Film Sensors Based on Fe<sub>2</sub>O<sub>3</sub>-SnO<sub>2</sub> Nanocomposites. *Sens. Actuators, B* **2004**, *101*, 199–206.

(46) Neri, G.; Bonavita, A.; Micali, G.; Rizzo, G.; Pinna, N.; Niederberger, M.; Ba, J. Effect of The Chemical Composition on The Sensing Properties of In<sub>2</sub>O<sub>3</sub>-SnO<sub>2</sub> Nanoparticles Synthesized by a Non-Aqueous Method. *Sens. Actuators, B* **2008**, *130*, 222–230.

(47) Khoang, N. D.; Trung, D. D.; Duy, N. V.; Hoa, N. D.; Hieu, N. V. Design of SnO<sub>2</sub>/ZnO Hierarchical Nanostructures for Enhanced Ethanol Gas-Sensing Performance. *Sens. Actuators, B* **2012**, *174*, 594–601.

(48) Ivanovskaya, M.; Kotsikau, D.; Faglia, G.; Nelli, P. Influence of Chemical Composition and Structural Factors of Fe<sub>2</sub>O<sub>3</sub>/In<sub>2</sub>O<sub>3</sub> Sensors on Their Selectivity and Sensitivity to Ethanol. *Sens. Actuators, B* **2003**, *96*, 498–503.

(49) Weis, T.; Lipperheide, R.; Wille, U. Barrier-Controlled Carrier Transport in Microcrystalline Semiconducting Materials: Description with a Unified Model. *J. Appl. Phys.* **2002**, *92*, 1411–1418.

(50) Zhou, W. W.; Cheng, C. W.; Liu, J. P.; Tay, Y. Y.; Jiang, J.; Jia, X. T.; Zhang, J. X.; Gong, H.; Hng, H. H.; Yu, T.; Fan, H. J. Epitaxial Growth of Branched α-Fe<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> Nano-Heterostructures with Improved Lithium-Ion Battery Performance. *Adv. Funct. Mater.* **2011**, *21*, 2439–2445.

(51) Gopal, R. C. B.; Cao, W.; Tan, O. K.; Zhu, W. Preparation of Fe<sub>2</sub>O<sub>3</sub>(0.9)-SnO<sub>2</sub>(0.1) by Hydrazine Method: Application as an Alcohol Sensor. *Sens. Actuators, B* **2002**, *81*, 170–175.

(52) Yin, L. W.; Li, M. S.; Bando, Y.; Golberg, D.; Yuan, X.; Sekiguchi, T. Tailoring The Optical Properties of Epitaxially Grown Biaxial ZnO/Ge, and Coaxial ZnO/Ge/ZnO and Ge/ZnO/Ge Heterostructures. *Adv. Funct. Mater.* **2007**, *17*, 270–276.

(53) Bao, M.; Chen, Y. J.; Li, F.; Ma, J. M.; Lv, T.; Tang, Y. J.; Chen, L. B.; Xu, Z.; Wang, T. H. Plate-like P-N Heterogeneous NiO/WO<sub>3</sub> Nanocomposites for High Performance Room Temperature NO<sub>2</sub> Sensors. *Nanoscale* **2014**, *6*, 4063–4066.

(54) Yamazoe, N.; Fuchigami, J.; Kishikawa, M.; Seiyama, T. Interactions of Tin Oxide Surface with O<sub>2</sub>, H<sub>2</sub>O and H<sub>2</sub>. *Surf. Sci.* **1979**, *86*, 335–344.

(55) Shimizu, Y.; Kuwano, N.; Hyodo, T.; Egashira, M. High H<sub>2</sub> Sensing Performance of Anodically Oxidized TiO<sub>2</sub> Film Contacted with Pd. *Sens. Actuators, B* **2002**, *83*, 195–201.